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LETTER TO THE EDITOR

Pressure-induced orthorhombic to rhombohedral phase transition in LaGaO₃

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Abstract

The influence of pressure on the crystal structure of $LaGaO_3$ has been studied using synchrotron x-ray diffraction. At atmospheric pressure $LaGaO_3$ has an orthorhombic structure in *Pnma* and undergoes a first-order phase transition to a rhombohedral structure near 2.5 GPa. The transition is reversible and the orthorhombic structure is recovered upon releasing the pressure below 2 GPa.

(Some figures in this article are in colour only in the electronic version)

1. Introduction

The perovskite lanthanum gallate (LaGaO₃) has lattice spacings that make it a good substrate for high- T_c superconductors (HTSC) and colossal magnetoresistive (CMR) film epitaxy [1–3]. Interest in this compound has been further fuelled by its promising behaviour as an electrolyte for solid state fuel cells [4–6]. For both applications it is typical to partially substitute for La and/or Ga in order to enhance its desirable properties.

Pure LaGaO₃ has an orthorhombic (*Pnma*) perovskite structure under ambient conditions and undergoes a first-order phase transition to a rhombohedral structure in $R\bar{3}c$ near 150 °C [7, 8]. These two structures differ with respect to the tilts of the GaO₆ octahedra. A description of the tilts in ABO₃ type perovskites has been given by Glazer [9] and the relationships between the possible tilt systems have recently been expanded upon by Howard and Stokes [10]. The *Pnma* to $R\bar{3}c$ transition in LaGaO₃ is sensitive to chemical substitution, and partially replacing the La with a smaller lanthanide ion such as Nd or Sm stabilizes the orthorhombic structure [11, 12]. The smaller lanthanide tends to increase the magnitude of the tilts of the octahedra. It is not surprising therefore that the temperature of the transition to the rhombohedral structure has been found to increase with Ln content in the three series La_{1-x}Ln_xGaO₃ Ln = Pr, Nd or Sm [11, 12]. There is now a considerable body of evidence



Figure 1. Observed and calculated room temperature x-ray powder diffraction profiles for LaGaO₃ at 1.10 and 4.80 GPa. The short vertical bars mark the positions of the allowed Bragg reflections. The broad feature near 20° is due to the gasket (marked with *) and was excluded from the calculation.

demonstrating that the tilts of the BO₆ octahedra are reduced with increasing temperature, and in numerous examples this leads to a sequence of phase transitions where the archetypal cubic (Pm3m) structure is ultimately obtained [13–18].

Pressure can be viewed as the third method of controlling the relative stability of the orthorhombic and rhombohedral phases. To a rough approximation the effect of pressure is expected to represent the inverse of temperature (increasing the pressure should reduce the volume of the cell as will reducing the temperature) and it can be postulated that phase transitions that occur upon cooling might be observed upon application of external pressure [19]. This was recently observed in another perovskite, PrAlO₃ [20]. This relationship between temperature and pressure assumes that the volume change at the phase transition is negligible, and the precise relationship between pressure and temperature has yet to be established. For LaGaO₃ the volume of the high-temperature rhombohedral phase is measurably smaller than that of the low-temperature orthorhombic phase [8] and the reverse might be expected, that is application of pressure may stabilize the rhombohedral phase at room temperature. Here we describe high-resolution x-ray diffraction studies of LaGaO₃ at pressures of up to 4.8 GPa aimed at examining this possibility.

2. Experiment

The polycrystalline sample of LaGaO₃ was prepared by the solid state reaction of stoichiometric amounts of La₂O₃ and Ga₂O₃ at 1173 K in air for 24 h and then, with periodic regrinding, at 1373 K for 48 h and 1623 K for 24 h. A modified Merrill–Bassett diamond-anvil cell was used for the x-ray studies in the pressure range 0.0–4.8 GPa. The powdered sample together



Figure 2. Pressure-induced evolution of a portion of the x-ray diffraction patterns for LaGaO₃ showing the conversion of the 020/112/200 triplet of the orthorhombic phase to the 110/014 doublet of the rhombohedral phase. The sharp peak near 14.75° in the pattern at 0.32 GPa is from a ruby. The pressures are, from the bottom 0.32, 1.10, 2.45, 3.08 and 4.80 GPa.

with a few small ruby chips was loaded into a 200 μ m hole in a steel gasket. A methanol– ethanol–water (16:3:1) mixture was used as the hydrostatic pressure transmitting medium. The pressure was calibrated by measuring the shift of the R₁ fluorescence line of ruby before and after each run. The x-ray diffraction data were collected using synchrotron radiation at the bending magnet beamline, X7A, of the National Synchrotron Light Source at Brookhaven National Laboratory. A wavelength of 0.7054 Å was selected using a bent Si 220 focusing crystal monochromator. Data were collected using a linear position-sensitive detector mounted about 450 mm from the sample. Further information on the configuration of the instrument and the data collection procedure can be found elsewhere [21]. The program RIETICA was employed to refine the lattice parameters using intensities extracted by the LaBail method [22].

3. Results and discussion

The ambient pressure diffraction pattern of LaGaO₃ at room temperature showed only reflections due to the perovskite-like phase. All observed reflections could be indexed on the basis of an orthorhombic cell with a = 5.5094(3), b = 5.4786(3) and c = 7.7532(5) Å. We were not able to refine the structure in the orthorhombic phase once the sample had been introduced into the diamond-anvil cell, presumably due to a combination of effects including preferred orientation, the limited 2θ angular range and overlap with extraneous peaks from the gasket and/or ruby. Nevertheless a LaBail type analysis proved straightforward and an example is shown in figure 1.

Figure 2 shows a portion of the x-ray diffraction patterns at room temperature collected with successively higher pressures. In orthorhombic *Pnma* symmetry the cubic 110 reflection



Figure 3. (a) Variation of the volume of $LaGaO_3$ and (b) of the lattice parameters with pressure. The open symbols were for samples studied with successively higher pressure and the closed symbols were collected as the pressure was progressively lowered from 4.8 GPa. The vertical line shows the approximate position of the orthorhombic to rhombohedral phase transition. In both cells the lattice parameters have been reduced to their equivalent cubic values.

is split into a triplet (the 020, 112 and 200 reflections) while in rhombohedral symmetry it appears as a doublet from the 110 and 014 reflections. Examination of the region of the diffraction pattern that includes these reflections shows that below 1.1 GPa the diffraction pattern consists of a triplet indicative of orthorhombic symmetry. Unfortunately the pattern at 0.3 GPa contains an additional reflection from the ruby, which also falls in this area. With increasing pressure the volume of the cell contracts (figure 3), and the peaks move to successively higher angles(figure 2). At 2.5 GPa the pattern in this region is essentially a doublet indicative of rhombohedral symmetry although some additional intensity is evident near $2\theta = 14.85^{\circ}$ between the rhombohedral 110 and 014 reflections, demonstrating that some residual orthorhombic phase persists. The existence of an extensive two-phase region has been noted in studies of the thermally induced transition in some substituted analogues of the type $La_{1-x}Ln_xGaO_3$, Ln = Sm, Pr, Nd [11, 12]. Above 2.5 GPa the pattern contains a simple doublet in this region, and as demonstrated in figure 1 the observed pattern is well reproduced in $R\overline{3}c$, this being the space group observed in LaGaO₃ in recent high-temperature powder neutron diffraction studies [7, 23].

Releasing the pressure reversed the transition with the pattern recorded at 1.83 GPa consisting predominantly of the orthorhombic phase together with a small amount of the rhombohedral phase. At pressures below this the pattern showed evidence for only the orthorhombic phase. Surprisingly there is very little hysteresis in the observed volume (figure 3), and upon total release the observed volume was very similar to that found for LaGaO₃ at room temperature.

The crystal structure of LaGaO₃ is on the border between the rhombohedral and orthorhombic structures. Transitions between these can be induced by increasing either the temperature or pressure or by adding strain to the lattice by chemical substitution. Obviously this is undesirable if LaGaO₃ is to be employed as a support for HTSC or CMR materials, although we are unaware of any studies where the structural phase transition has been shown to be a factor in a device. It is clear that chemical substitutions can increase the transition temperature, and it remains to be determined if these too increase the pressure needed to induce the transition to the rhombohedral structure. Determining the precise relationship between temperature and pressure in phase transitions of these types is an obvious priority.

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References

- [1] Koren G, Gupta A, Giess E A, Segmuller A and Laibowitz R B 1989 Appl. Phys. Lett. 54 1054
- [2] Sandstrom R L, Giess E A, Gallagherm W J, Segmuller A, Copper E I, Chisholm M F, Gupta A, Shinde S and Laibowitz R B 1988 Appl. Phys. Lett. 53 1087
- [3] Kawasaki M, Izumi M, Konishi Y, Manako T and Tokura Y 1999 Mater. Sci. Eng. B 63 49
- [4] Ishihara T, Matsuda H, bin Bustam M A and Takita Y 1996 Solid State Ion. 86-8 197
- [5] Kharton V V, Viskup A P, Naumovich E N and Lapchuk N M 1997 Solid State Ion. 104 67
- [6] Ismihara T, Honda M, Shibayama T, Minami H, Nishiguchi H and Takita Y 1998 J. Electrochem. Soc. 145 3177
- [7] Howard C J and Kennedy B J 1999 J. Phys.: Condens. Matter 11 3229
- [8] Wang Y, Liu X, Yao G D, Liebermann R C and Dudley M 1991 Mater. Sci. Eng. A 132 13
- [9] Glazer A M 1972 Acta Crystallogr. 28 3384
- [10] Howard C J and Stokes H 1998 Acta Crystallogr. 54 782
- [11] Vasylechko L, Niewa R, Borrmann H, Knapp M, Savytskii D, Matkovski A, Bismayer U and Berkowski M 2001 Solid State Ion. 143 219

- [12] Berkowski M et al 2000 J. Crystal Growth 209 75
- [13] Knight K S 1994 Solid State Ion. 74 109
- [14] Kennedy B J and Hunter B A 1998 Phys. Rev. B 58 653
- [15] Darlington C N W and Knight K S 1999 Acta Crystallogr. 55 24
- [16] Kennedy B J, Howard C J and Chakoumakos B C 1999 J. Phys.: Condens. Matter 11 1479
- [17] Kennedy B J, Prodjostanoso A K and Howard C J 1999 J. Phys.: Condens. Matter 11 6319
- [18] Howard C J, Knight K S, Kennedy B J and Kisi E H 2000 J. Phys.: Condens. Matter 12 L677
- [19] Hazen R M and Finger L W 1982 Comparative Crystal Chemistry (New York: Wiley)
- [20] Kennedy B J, Vogt T, Martin C D, Parise J B and Hriljac J A at press
- [21] Jephcoat A P, Finger L W and Cox D E 1992 High Press. Res. 8 667
- [22] Howard C J and Hunter B A 1998 A Computer Program for Rietveld Analysis of X-ray and Neutron Powder Diffraction Patterns (NSW, Australia: Lucas Heights Research Laboratories)
- [23] Lerch M, Boysen H and Hansen T 2001 J. Phys. Chem. Solids 62 445